We present a theoretical study of photocarrier dynamics in graphene due to electron-phonon (EP) interactions. Using the relaxation-time approximation (RTA) with parameters determined from density functional theory (DFT) and a complementary explicitly solvable model, we show that the photocarrier thermalization time changes by orders of magnitude, when the excitation energy is reduced from 1 eV to the 100 meV range. In detail the ultrafast thermalization at low temperatures takes place on a femtosecond timescale via optical phonon emission, but slows down to picoseconds once excitation energies become comparable with these optical phonon energy quanta. In the latter regime thermalization times exhibit a pronounced dependence on temperature. Our DFT-based model includes all the inter- and intraband transitions due to EP scattering. Thanks to the high melting point of graphene we extend our studies up to 2000 K and show that such high temperatures reduce the photocarrier thermalization time through phonon absorption.

DOI: 10.1103/PhysRevB.99.155410

I. INTRODUCTION

Recent progress in nanotechnology has made it possible to fabricate high-quality materials that are only one atom thick and hence reach the fundamental two-dimensional (2D) limit for solid crystals [1]. Due to their ultimate thinness these materials demonstrate various properties that are qualitatively different from those of the three-dimensional parent crystals and, at the same time, are found to be useful in photodetection and photovoltaic applications [2]. Indeed, the central phenomenon employed in photodetection and photovoltaics is the conversion of light energy into electricity. It is a quantum conversion process, employing absorption of photons to deliver photoexcited carriers to an external circuit, where they do electrical work [3]. There are two obvious strategies for increasing the amount of energy transferred by photocarriers. One can try to speed up the photocarrier extraction such that the carriers are collected, while they are still hot or even out of thermal equilibrium. Alternatively, one can try to slow down the cooling or photocarrier thermalization for the same purpose.

Graphene in a combination with other 2D semiconductors offers an interesting opportunity to employ both strategies. Thanks to the extremely small thickness of the junctions between 2D materials (also known as van der Waals heterostructures [4]), interlayer photocarrier transport may occur faster than the intralayer relaxation processes [5]. At the same time the optical phonon emission is strongly suppressed for low-energy excitations in graphene due to unusually high energy quanta of optical phonons [6,7]. As a consequence, the photocarriers can be extracted well before they thermalize and dissipate useful energy by means of phonon emission. By incorporating graphene into a heterostructure, we can combine the two strategies in one optoelectronic device. In this way the photoresponse can be substantially increased simultaneously to the device performance. In this paper we focus on the photocarrier evolution in intrinsic graphene, providing conclusive evidence for the existence of a thermalization bottleneck that makes such applications possible.

The striking difference between electron thermalization in intrinsic graphene and in a metal (e.g., doped graphene) is that the electron-electron collision rate is much lower in the former than in the latter, as it has been pointed out and confirmed by means of Raman spectroscopy by Basko et al. [8]. Indeed two electrons colliding with each other have to satisfy energy and momentum conservation. If one electron is excited in the conduction band, then it most likely collides with another electron from the valence band (interband electron-electron scattering), as the conduction band is almost empty in intrinsic graphene. The only possible way to satisfy both conservation laws in this case is to assume all four involved momenta to be parallel (collinear scattering). This is a direct consequence of the conelike dispersion for carriers in graphene. Other collinear channels involved in the electron-electron scattering are Auger recombination and carrier multiplication. Both lead to a change in the excited charge carrier density [9–14]. Carrier multiplication is favorable at low pump fluence and high excitation energies, whereas Auger recombination requires high pump fluence and low excitation energies [9,11,15]. Furthermore, these processes are very sensitive to the doping level and temperature [13–16]. Collinear scattering is an inefficient thermalization mechanism, and the corresponding electron-electron thermalization time is about 1 ps for photocarriers excited in the optical spectral region, and it is even longer for lower excitation energies [6,7,17]. Since our model is developed to provide a solid proof for the photocarrier thermalization bottleneck in intrinsic graphene that occurs at the excitation energy near 100 meV, the electron-electron scattering turns out to be irrelevant for such excitation energies and only EP scattering is taken into account. This assumption is...
invalid for conventional heavily doped graphene samples with a Fermi energy of about 200 meV. However, such samples would not be suitable for studying the photocarrier thermalization bottleneck, as the relevant interband excitations are blocked by the Pauli principle.

In contrast to advanced two-temperature models for electron thermalization in metals [18–20] and semiconductors [21] we are dealing here with a single temperature describing both the thermalized electrons and equilibrium phonons. This single-temperature model does not treat the thermalization of phonons, which would require us to take into account additional effects such as phonon-phonon scattering due to anharmonicities [22]. Neglecting weak perturbations from nonequilibrium phonon occupations, the photoexcited electrons in our model relax directly to the Fermi-Dirac distribution that is already in thermal equilibrium with the phonon bath. The temperature can be much higher than 300 K due to sample heating, but we assume the absorbed radiation fluence to be low enough to prevent a further increase of this temperature.

The photocarrier dynamics in graphene has been studied experimentally by means of pump-probe spectroscopy as well as time- and angle-resolved photoemission spectroscopy [23–31]. In the experiments the carriers are excited far above the Dirac point (by more than 1 eV), and the ultrafast relaxation of hot carriers is mainly attributed to optical phonon emission and electron-electron scattering, taking place within 150–170 fs [23,28]. Excitations below the highest optical phonon frequency of graphene through density functional perturbation theory for electronic and phononic properties

We use DFT within the local density approximation (LDA) to calculate the ground-state electronic properties of graphene with QUANTUM ESPRESSO [47]. We employ a plane-wave basis set with a kinetic energy cutoff of 110 Ry, a charge density cutoff of 440 Ry and a Troullier-Martins pseudopotential for carbon with a $2s^22p^2$ valence configuration [48]. The unit cell of graphene is relaxed with the help of the Broyden-Fletcher-Goldfarb-Shanno algorithm until the net force on atoms is less than $10^{-6}$ Ry/a.u., and total energy changes are below $10^{-8}$ Ry. A vacuum of 20 Å along the out-of-plane direction is used to avoid artificial interactions with periodic images of the graphene sheet, and the BZ is sampled with a $45\times45\times1$ Γ-centered $k$ grid. We construct Wannier functions to get localized orbitals from plane-wave eigenfunctions. By interpolating wave functions, we finally obtain electronic eigenenergies, dynamical matrices, and EP couplings on fine grids in the BZ [49]. We calculate the phonon dispersion spectrum of graphene through density functional perturbation theory (DFPT) [50], employing a $12\times12\times1$ $q$ grid to evaluate phonon dynamical matrices.

By performing the DFT procedures, we obtain an optimized in-plane lattice constant of graphene of $a = |a_1| = |a_2| = 2.436$ Å, see Fig. 1(a), which is in good agreement with previous reports of 2.458 Å [51]. We calculate electronic and phononic band structures along high symmetry lines of the first BZ, as plotted in Fig. 1(b). Figure 1(c) shows the electronic band structure, as computed from DFT with plane waves. The excellent agreement with those determined through the Wannier function method demonstrates the high quality of the interpolated localized orbitals. The phonon dispersion is finally displayed in Fig. 1(d).

Having determined electronic and phononic band structures, we calculate the electronic self-energy $\Sigma_{nk}(T)$ due to the EP interaction for the electronic eigenstate $|nk\rangle$ with the EPW code. It is defined as follows [52]:

$$
\Sigma_{nk}(T) = \sum_{m,p} \int_{BZ} \frac{d^3q}{2q_{BZ}} |g_{mn,p}(\mathbf{k}, \mathbf{q})|^2 \left[ \frac{N_{\omega q_{nk}}(T) + f_{\omega q_{nk}}^{(0)}(T)}{\epsilon_{nk} - (\epsilon_{nk+q} - \epsilon_F) + i\eta} \right. \\
\left. + \frac{N_{\omega q_{nk}}(T) + 1 - f_{\omega q_{nk}}^{(0)}(T)}{\epsilon_{nk} - (\epsilon_{nk+q} - \epsilon_F) + i\eta} \right],
$$

Our paper is organized as follows. In Sec. II we present the theoretical approaches used in this work. Next, we discuss the results obtained within the models in Sec. III before we end with a summary and outlook in Sec. IV.
dynamical matrices do not change with EP interactions [52]. The first grid, which we find necessary to accurately map out the whole [53]. We use a 1200×36×1 k grid using Wannier functions [54].

eigenenergies, wave functions, and phonon dynamical matrices are interpolated on fine grids using Wannier functions [55].

eigenenergies, wave functions, and phonon dynamical matrices are interpolated on fine grids using Wannier functions [55].

and momentum, is calculated as

\[
\tau_{nk}(T) = \frac{\hbar}{2\text{Im}[\Sigma_{nk}(T)]}
\]

from Eq. (1) by using the imaginary part of the self-energy.

B. Time evolution of excited charge carriers

The time evolution of the electronic occupation \( \tilde{f}_{nk}(t, T) \) is calculated using the Boltzmann equation in the RTA

\[
\frac{d\tilde{f}_{nk}(t, T)}{dt} = -\frac{\tilde{f}_{nk}(t, T) - f^{(0)}_{nk}(T)}{\tau_{nk}}
\]

with the solution

\[
\tilde{f}_{nk}(t, T) = f^{(0)}_{nk}(T) + e^{-\frac{t}{\tau_{nk}}} \left[ \tilde{f}_{nk}(0, T) - f^{(0)}_{nk}(T) \right].
\]

if the excitation is assumed to happen at time \( t = 0 \). Equation (5) states that when the system is weakly perturbed, the perturbation decays exponentially with the scattering time \( \tau_{nk} \) to restore the equilibrium Fermi-Dirac distribution \( f^{(0)}_{nk}(T) \) [54]. RTA omits all the intermediate relaxation steps between an initial nonequilibrium electron distribution and its final relaxed occupation so that an excited electron relaxes directly to the thermalized state. It is valid only when the system is weakly perturbed from the equilibrium and higher order perturbations can be safely neglected. We consider here the isotropic scattering processes only, i.e., there is no angular dependence and \( \tau_{nk} \) is the average time between the EP collision events. The tilde sign indicates the time dependence of the occupation function.

We generate the initial hot-carrier occupation \( \tilde{f}_{nk}(0, T) \) as a combination of a Fermi-Dirac distribution \( f^{(0)}_{nk}(T) \) at the temperature \( T \) and a Gaussian peak at energy \( +\zeta \) for electrons in the conduction band (\( \epsilon_{nk} > \epsilon_F \)) and \( -\zeta \) for the holes in the valence band (\( \epsilon_{nk} < \epsilon_F \)) as

\[
\tilde{f}_{nk}(0, T) = f^{(0)}_{nk}(T) \left\{ \begin{array}{ll}
\frac{-\epsilon_{nk} + \epsilon_F}{\sqrt{2\pi}\sigma} e^{-\frac{\epsilon_{nk}^2}{2\sigma^2}}, & \epsilon_{nk} \geq \epsilon_F, \\
\frac{\epsilon_{nk} - \epsilon_F}{\sqrt{2\pi}\sigma} e^{-\frac{\epsilon_{nk}^2}{2\sigma^2}}, & \epsilon_{nk} < \epsilon_F.
\end{array} \right.
\]

Throughout this work we choose a small energy smearing \( \sigma = 8.47 \text{ meV} \) and small perturbation \( \lambda_c = 2.4\times10^{-3} \text{ eV} \). The parameter \( \lambda_h \) is selected such that the initially excited number of electrons and holes is the same. Since the density of states (DOS) of graphene is rather symmetric in the range of excitation energies \( -0.8 \lesssim \zeta \lesssim 0.8 \text{ eV} \) studied by us [see Fig. 1(c)], it turns out to be an excellent approximation to set \( \zeta = \lambda_c = \lambda_h \).

While we use \( \lambda \) here as a free parameter to adjust the initial occupation, it can be related to measurements through \( \lambda = 4\pi^2\alpha e^2h^2\Phi v_F^2/\zeta^2 \). In the expression, \( \alpha \) is the linear absorption of graphene, \( \Phi \) is the pump-fluence, and \( v_F \) is the Fermi velocity of electrons in graphene [17].

We determine the time \( \tau_{nh} \) when hot carriers have relaxed through the relation \( P(\zeta, 0, T) - P(\zeta, t, T) < P(\zeta, 0, T)/e \). In the expression we have defined the population

\[
P(E, t, T) = \sum_{nk} \delta(E - \epsilon_{nk}) \left\{ \begin{array}{ll}
1 - \tilde{f}_{nk}(t, T), & E < \epsilon_F, \\
\tilde{f}_{nk}(t, T), & E \geq \epsilon_F.
\end{array} \right.
\]
Our definition ensures that the population is symmetric with regard to electrons and holes, as long as the DOS is symmetric.

### C. Analytical model

Before performing parameter-free \textit{ab-initio} calculations of charge carrier dynamics, we estimate the photocarrier thermalization time of intrinsic graphene within an explicitly solvable model. For simplicity we assume only optical phonon modes \( p \) that are dispersionless, i.e., exhibit the fixed energy \( \hbar \omega_p \). For this reason phonon wave vectors will be omitted. Furthermore, we consider only two linear electronic bands of the Dirac cone with \( \varepsilon_{\pi \hbar} = n \hbar v_F k, n = \pm \) and \( k = |k| \). Additionally, we will suppress all time and temperature arguments of the occupation functions in this subsection, because we assume a steady state regime.

The EP collisions in the given optical phonon mode \( p \) are governed by the following integral:

\[
I_p[f_{\pi k}] = \sum_{\pi' k'} [f_{\pi' k'} (1 - \tilde{f}_{\pi k}) W_{\pi' k' \rightarrow \pi k} - \tilde{f}_{\pi' k'} (1 - \tilde{f}_{\pi k}) W_{\pi k \rightarrow \pi' k'}],
\]

where \( \tilde{f}_{\pi k} = f^{(0)}_{\pi k} + f^{(1)}_{\pi k} \) denotes the carrier occupation with the time-independent Fermi-Dirac distribution \( f^{(0)}_{\pi k} \) and the nonequilibrium addition \( f^{(1)}_{\pi k} \), representing the second term in Eq. (5). The transition probability is given by Fermi’s golden rule

\[
W_{\pi k \rightarrow \pi' k'} = \frac{2 \pi}{\hbar} W_p [(N_p + 1) \delta(\varepsilon_{\pi' k'} - \varepsilon_{\pi k} - \hbar \omega_p) + N_p \delta(\varepsilon_{\pi' k'} - \varepsilon_{\pi k} + \hbar \omega_p)]
\]

for carriers outgoing from the state \( |\pi k\rangle \), and

\[
W_{\pi k' \rightarrow \pi k} = \frac{2 \pi}{\hbar} W_p [(N_p + 1) \delta(\varepsilon_{\pi k'} - \varepsilon_{\pi k} - \hbar \omega_p) + N_p \delta(\varepsilon_{\pi k'} - \varepsilon_{\pi k} + \hbar \omega_p)]
\]

for carriers incoming to the state \( |\pi' k\rangle \). Making use of nearly dispersionless optical phonon modes, the EP interaction matrix element \( W_p \) is assumed to be independent of momentum. The first term in both Eqs. (9) and (10) corresponds to the phonon emission, while the second one describes the phonon absorption. The phonons are treated as a noninteracting gas, characterized by the Bose-Einstein distribution \( N_p = N_0 e^{-\omega/\hbar T} \). Due to the strong carbon-carbon bonding in graphene the optical phonon energy is higher than 100 meV [see Fig. 1(c)] and, hence, we assume \( \hbar \omega_p \gg k_B T \) for typical temperatures or, in other words, \( N_p \ll 1 \). The collision integral can then be simplified to

\[
I_p[f_{\pi k}] = \frac{2 \pi}{\hbar} W_p \sum_{\pi' k'} [f_{\pi' k'} (1 - \tilde{f}_{\pi k}) \delta(\varepsilon_{\pi' k'} - \varepsilon_{\pi k} - \hbar \omega_p) - \tilde{f}_{\pi' k'} (1 - \tilde{f}_{\pi k}) \delta(\varepsilon_{\pi k} - \varepsilon_{\pi' k'} - \hbar \omega_p)].
\]

Let us now assume \( \tilde{f}_{\pi k} \) to be a function of \( \varepsilon_{\pi k} \) and integrate in momentum space. Making use of the \( \delta \) function and \( \varepsilon_{\pi k} = \varepsilon_{\pi k} \), we obtain

\[
I_p[f_{\pi k}] = \frac{W_p}{\hbar^2 v_F^2} \left[ \varepsilon_{\pi k} + \hbar \omega_p f_{\pi k} + \hbar \omega_p (1 - f_{\pi k}) - |\varepsilon_{\pi k} - \hbar \omega_p| f_{\pi k} (1 - f_{\pi k} - \hbar \omega_p) \right].
\]

Finally, we employ a linear response approximation and the property of intrinsic graphene \( 1 - f^{(0)}_{\pi k} = f^{(0)}_{-\pi k} \) so that

\[
\tilde{f}_{\pi k} + \hbar \omega_p (1 - f^{(0)}_{\pi k}) \approx f^{(0)}_{\pi k} + \hbar \omega_p f^{(0)}_{-\pi k} - f^{(1)}_{\pi k} f^{(0)}_{\pi k} + \tilde{f}^{(1)}_{\pi k} f^{(0)}_{\pi k} + f^{(1)}_{\pi k} f^{(0)}_{\pi k} - \tilde{f}^{(1)}_{\pi k} f^{(0)}_{-\pi k} - f^{(1)}_{\pi k} f^{(0)}_{-\pi k},
\]

\[
\tilde{f}_{\pi k} (1 - \tilde{f}_{\pi k} - \hbar \omega_p) \approx f^{(0)}_{\pi k} - f^{(0)}_{\pi k} + \tilde{f}^{(1)}_{\pi k} f^{(0)}_{\pi k} - f^{(1)}_{\pi k} f^{(0)}_{\pi k} - \tilde{f}^{(1)}_{\pi k} f^{(0)}_{-\pi k} + f^{(1)}_{\pi k} f^{(0)}_{-\pi k}.
\]

Hence, Eq. (12) can be written as a sum of two terms \( I_p[f^{(0)}_{\pi k}] + I_p[f^{(0)}_{\pi k} f^{(1)}_{\pi k}] \), where

\[
I_p[f^{(0)}_{\pi k} f^{(1)}_{\pi k}] = \frac{W_p}{\hbar^2 v_F^2} \left[ \varepsilon_{\pi k} + \hbar \omega_p f^{(0)}_{\pi k} + \hbar \omega_p f^{(0)}_{\pi k} - |\varepsilon_{\pi k} - \hbar \omega_p| f^{(1)}_{\pi k} f^{(0)}_{\pi k} \right],
\]

\[
I_p[f^{(0)}_{\pi k} f^{(1)}_{\pi k}] = \frac{W_p}{\hbar^2 v_F^2} \left[ |\varepsilon_{\pi k} + \hbar \omega_p| f^{(0)}_{\pi k} + \hbar \omega_p f^{(0)}_{\pi k} - \varepsilon_{\pi k} - \hbar \omega_p| f^{(1)}_{\pi k} f^{(0)}_{\pi k} \right].
\]

Equations (15) and (16) are valid for any ratio between \( \varepsilon_{\pi k} \) and \( \hbar \omega_p \) so that we can investigate the thermalization behavior for photocarriers excited below and above the phonon frequency. Note that only Eq. (16) is responsible for thermalization, because Eq. (15) does not contain \( f^{(1)}_{\pi k} \).

In what follows we consider the thermalization of electrons (i.e., \( \varepsilon_k = \varepsilon_k + \hbar \omega_F k \)), as the thermalization of holes is equivalent in the case of intrinsic graphene at not too high excitation energies [see Fig. 1(c)]. Assuming the initial nonequilibrium distribution to be \( \delta \) shaped, \( f^{(1)}_{\pi k} \propto \delta(\varepsilon_k - \hbar \omega/2) \), we find

\[
I_p[f^{(1)}_{\pi k}] = \frac{\omega W_p}{2 \hbar^2 v_F^2} \left( f^{(1)}_{\pi k} f^{(0)}_{\pi k} + \tilde{f}^{(1)}_{\pi k} f^{(0)}_{\pi k} + f^{(1)}_{\pi k} f^{(0)}_{\pi k} - \tilde{f}^{(1)}_{\pi k} f^{(0)}_{\pi k} + f^{(1)}_{\pi k} f^{(0)}_{\pi k} - \tilde{f}^{(1)}_{\pi k} f^{(0)}_{\pi k} \right).
\]
point at $E = \varepsilon_F = 0$, it follows the same behavior as the electronic DOS [see Eq. (1)]. This results from the fact that the electronic DOS represents the phase space for EP scattering events to take place. As can be inferred from Figs. 2(a) and 2(b), $\text{Im}[\Sigma_{nk}(T)]$ is very sensitive to temperature close to $E = 0$. In contrast it shows a much weaker temperature dependence at energies above around 200 meV, coinciding with the highest optical phonon energies. Indeed, we see for low temperatures (0–300 K) that $\text{Im}[\Sigma_{nk}]$ increases roughly exponentially until the highest optical phonon energy is reached, while the energy dependence is comparatively weak for elevated temperatures (600–2000 K). The behavior shows that scattering below the optical phonon threshold takes place rather inefficiently via acoustical phonons. With increasing temperature there are more phonons available for the carriers to interact with, leading to the increase of $\text{Im}[\Sigma_{nk}(T)]$. Analogously, the available phase space for optical phonon emission grows with increasing energy.

In the inset of Fig. 2(b) we consider the scattering times $\tau_{nk}(T)$, which are inversely proportional to the self-energy [see Eq. (3)]. We observe that around the Dirac point the scattering time becomes very sensitive to temperature and can be on the order of a few picoseconds for low $T$. In contrast, at energies above 200 meV the scattering times exhibit only weak energy and temperature dependencies. As argued before, this behavior can be rationalized by the fact that for low $T$ at $E < 200$ meV excited carriers can relax via acoustical phonon scattering only, while they thermalize efficiently via optical phonons above 200 meV.

The behavior of $\text{Im}[\Sigma_{nk}(T)]$ in Fig. 2 can also be analyzed in terms of Eq. (1). Let's consider low temperatures and electrons with $\varepsilon_{nk} \geq 0$. In this case both $f_{nk}^{(0)}(T)$ and $N_{\text{hot}}(T)$ are vanishingly small, and thus only the second term of Eq. (1) contributes. For this reason, excited electrons relax via emission of phonons. But as temperature increases, we get $0 \lesssim f_{nk}^{(0)}(T) \lesssim 1$ and $N_{\text{hot}}(T) > 0$, and both terms in Eq. (1) start contributing. For this reason $\text{Im}[\Sigma_{nk}(T)]$ increases with increasing temperature in Fig. 2 for $E > 0$. An analogous argumentation can be carried out for holes.

To simulate the temporal dynamics, we use Eq. (5), starting with the initial distribution of Eq. (6) at time $t = 0$. Choosing the parameters $\kappa$ and $\sigma$ as described above, we calculate time evolutions of occupations for different temperatures $T$ and excitation energies $\zeta$. We are particularly interested in the behavior of thermalization times for excitations below and above the optical phonon threshold.

Figure 3 shows the hot carrier population $P(E, t, T)$ [see Eq. (7)] for excitation energies $\zeta = 0.05$, 0.5 eV and temperatures $T = 0$, 10, 100 K. Below the optical phonon threshold for $\zeta = 0.05$ eV in Figs. 3(a)–3(c), thermalization of the hot carriers takes place on the picosecond timescale via low-energy acoustical phonons. In this excitation range the relaxation time decreases with increasing temperature, because the background equilibrium electron distribution allows excited carriers to scatter increasingly efficiently with the optical phonons [32]. Our thermalization time $\tau_{th}$ at $T = 10$ K, as extracted from Fig. 3(b), is around 175 ps. This is lower than the 300 ps reported in Ref. [32] for an excitation energy of 51 meV on an epitaxially grown graphene sample containing around ~70 layers and arranged over a SiC substrate. Above
the optical phonon threshold, our results in Figs. 3(d)–3(f) predict a weak or almost no temperature dependence of the relaxation time. With \( \tau_{th} \approx 60 \text{ fs} \) it takes a value of similar size as the photocarrier isotropization time from Ref. [31], originating from scattering by optical phonons. Our qualitative findings of a strong temperature dependence of \( \tau_{th} \) below the optical phonon threshold and none above are consistent with the experimental observations in Ref. [32]. The plots in Fig. 3 also demonstrate that the populations of electrons and holes evolve with time quite symmetrically around the Dirac point, confirming that the dynamics of holes are similar to those of electrons.

Due to the extraordinarily high melting temperature of nearly 5000 K predicted theoretically for graphene [46], we extend our analysis of time evolutions to high temperatures \( T = 300, 600, 1200, 2000 \text{ K} \), see Fig. 4. We find carriers to relax at \( T = 300 \) or \( 600 \text{ K} \) on a 100 fs time scale. At 1200 K this reduces to around 34 fs and is even below 26 fs at 2000 K.

In Fig. 5 we summarize the relaxation times \( \tau_{th} \), which we have extracted from the Boltzmann equation combined with our \textit{ab-initio} modeling at different excitation energies and temperatures. For \( \zeta = 0.05 \text{ eV} \) the thermalization time decreases with increasing temperature from \( T = 0 \) to 1200 K by more than three orders of magnitude. In contrast, there is only little change in the relaxation time with temperature for a fixed excitation with \( \zeta = 0.4, 0.6, 0.8 \text{ eV} \) above the optical phonon threshold. A slight decrease is seen at the temperatures, where thermal energies are similar to those of optical phonon quanta, i.e., \( k_{B}T \approx \hbar \omega_{p} \). In addition, for a fixed temperature, relaxation times depend only little on \( \zeta \), if the excitation energy is above the optical phonon threshold.

To summarize, taking into account only EP scattering events, we thus observe intriguingly that relaxation times in graphene can span an extraordinary range from 170 ps down to 60 fs, if the temperature is varied and carriers are excited below the optical phonon threshold.

Our predictions can be qualitatively understood by using the concept of a thermalization bottleneck in graphene. Thanks to the high optical phonon energy quanta of about 200 meV [see Fig. 1(d)], the low-energy (THz) electrons cannot relax as fast as the optically excited photocarriers, because at low temperatures (i) the phonon absorption is a very rare process and (ii) the phonon emission requires an empty electron state below the Fermi level, but states below \( \epsilon_{F} \) are almost fully occupied. The relevant thermalization times can be estimated by using our analytical model. We assume
an explicit form for the EP interaction matrix element given by [36]

\[ W_p = \frac{\hbar \Delta_p^2 F_p}{2 \rho \omega_p} \]  

(21)

where \( \Delta_p \) is the deformation potential for a mode \( p \), \( F_p \) is a dimensionless geometric factor, and \( \rho = 7.6 \times 10^{-8} \text{g/cm}^2 \) is the mass density. In what follows, we take into account the two most important phonon modes [36], \( \rho = \Gamma, K \), where \( F_{\Gamma} = 1, F_{K} = 1/2 \), \( \hbar \omega_{\Gamma} = 197 \text{meV} \), \( \hbar \omega_{K} = 157 \text{meV} \), and \( \Delta_\Gamma = \Delta_0, \Delta_K = \sqrt{2} \Delta_0 \) with \( \Delta_0 = 11 \text{eV/Å} \) [36]. At \( \omega \gg \omega_p \), the thermalization time can be found from Eq. (19) as

\[ \tau_{th} = \frac{4 \omega_0 \hbar v_F^2 \rho}{\omega \Delta_0^2}, \quad \omega \gg \omega_{\Gamma,K}. \]  

(22)

where \( 1/\omega_0 = 1/\omega_{\Gamma} + 1/\omega_{K} \). Assuming an excitation energy of \( \hbar \omega = 1.35 \text{eV} \) (i.e., a radiation wavelength of 800 nm), we estimate \( \tau_{th} \approx 58 \text{fs} \). In the opposite limit \( \omega \ll \omega_p \) we get from Eq. (20)

\[ \tau_{th} = \frac{\hbar v_F^2 \rho}{\Delta_0^2} \exp\left(-\frac{\hbar \omega_0}{\omega}\right) + \exp\left(-\frac{\hbar \omega_{K}}{\omega}\right), \quad \omega \ll \omega_{\Gamma,K}. \]  

(23)

Assuming the most relevant temperature of 300 K, we estimate \( \tau_{th} \approx 92 \text{ps} \).

Our considerations confirm that (i) the thermalization timescales differ at \( \omega \ll \omega_p \) and \( \omega \gg \omega_p \) by three orders of magnitude at room temperature, (ii) the photocarrier thermalization time strongly depends on temperature at \( \omega \ll \omega_p \), whereas at \( \omega \gg \omega_p \) it does not, and (iii) in the former case, the thermalization time decreases rapidly with increasing temperature. This is exactly what we see in the summary of the relaxation times \( \tau_{th} \) shown in Fig. 5, as determined through our first-principles approach.

**IV. SUMMARY AND OUTLOOK**

In summary, we have studied the relaxation dynamics of hot carriers in single-layer graphene near and away from the Dirac point subject to the EP interaction. By determining electron and phonon dispersions as well as EP couplings from DFT, our model based on the Boltzmann equation in the RTA contains no free parameters and takes into account contributions from all of the optical as well as acoustical branches in the whole BZ along with the inter- and intraband transitions taking place. In excellent agreement with analytical predictions we find that relaxation times computed with our model are strongly increased, if carriers are excited below the optical phonon energies. In addition, we have shown that the carrier relaxation times depend strongly on temperature for such low excitation energies, while being rather temperature-independent for excitation energies above optical phonon energy quanta. Our results are in agreement with the experimental findings, attributing photocarrier relaxation at different excitation energies to different phonon branches [23,28,32].

DFT results depend on the choice of the exchange-correlation functional. To assess the robustness of DFT-derived electron and phonon dispersions as well as EP couplings, different functionals have been tested in Ref. [55] and compared to the Hartree-Fock and GW approximations. While the conelike electron dispersion at the K point is consistently reproduced and phonon dispersions are generally found to be in good agreement with experiment, inaccuracies of the EP couplings for the highest optical phonon branch near the K point have been corroborated and the GW approach has been advocated as a way for making quantitative improvements. While the use of electronic structure methods beyond DFT would certainly be desirable in the future, we expect that more accurate ab-initio input parameters will not qualitatively change the results presented here.

The photocarrier thermalization bottleneck could be employed to facilitate the photoexcited electron transport from graphene to a semiconductor across a Schottky barrier [56–58]. Thanks to the longer relaxation time at lower excitation energies, the photocarriers can contribute to the interlayer transport before thermalization is completed, thus improving the photoresponsivity [59]. From the device engineering point of view, the most important assumption made in this work is the absence of a substrate. It might provide additional dielectric screening and unintentional doping, which overall influence the electron-electron scattering contribution neglected here. Moreover, the photocarriers might experience interactions with remote polar surface phonons [36], providing additional EP scattering channels. Since the precise effects caused by a substrate strongly depend on the chosen material and its interface properties, the model should be tailored for each device to make quantitative predictions. Such aspects could be addressed in future work.

**ACKNOWLEDGMENTS**

D.Y. and F.P. acknowledge financial support from the Carl Zeiss Foundation as well as the Collaborative Research Center (SFB) 767 of the German Research Foundation (DFG). M.T. is supported by the Director’s Senior Research Fellowship from the Centre for Advanced 2D Materials at the National University of Singapore (NRF Medium Sized Centre Programme R-723-000-001-281) and thanks the Okinawa Institute of Science and Technology for its hospitality. Part of the numerical modeling was performed using the computational resources of the bwHPC program, namely the bwUniCluster and the JUSTUS HPC facility.

---


